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Complete Characterization of Thermoelectric Materials by Impedance Spectroscopy

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ABSTRACT

Thermoelectric materials can directly convert waste heat into electricity. Due to the vast amount of energy available as waste heat in our society, these materials could contribute to reduce our dependence on fossil fuels and their associated environmental problems. However, the heat to electricity conversion efficiency of thermoelectric materials is still a limiting factor, and extensive efforts are being undertaken to improve their performance. The search for more efficient materials is focused on the optimization of three properties (Seebeck coefficient, electrical resistivity, and thermal conductivity). Typically, these are determined as function of temperature through independent measurements on two or more instruments, making thermoelectric characterization tedious and time consuming, which complicates the attainment of a more efficient heat to electricity energy conversion. Here, it is demonstrated for the first time that a complete thermoelectric characterization of a material may be achieved from a single electrical measurement performed on one instrument only, by employing the impedance spectroscopy method. A skutterudite sample is used for the demonstration, which is sandwiched between two stainless steel contacts in a four-probe arrangement and their properties are determined from 50 to 250 °C. This new approach shows good precision and agrees with characterization of the same sample performed with commercial equipment, illustrating the power of the technique to facilitate the rapid and efficient evaluation of thermoelectric materials.

1. INTRODUCTION

Nowadays, more than 60% of the global power is lost as waste heat, which represents ≈ 15 TW. A 10% recovery of this energy will exceed the summation of most current renewable energy sources (solar, wind, geothermal, and hydro energy).¹ Thermoelectric (TE) materials can directly convert waste heat into electricity. Due to this, they have interest in applications such as automobiles and industries, where they can generate energy from the waste heat released by exhaust gases and reduce CO₂ emissions.² They can also convert solar warmth into electricity when integrated in solar thermoelectric generators.³ In addition, they are also potentially able to power wearable electronics and sensors using environmental heat or from human bodies, being a top candidate for self-powering sensors from the internet of things, empowering the elimination of batteries, which are toxic and subjected to frequent recharging and replacement.⁴ An efficient heat to electricity energy conversion from these applications would help to reduce our dependence on fossil fuels and their associated environmental problems.

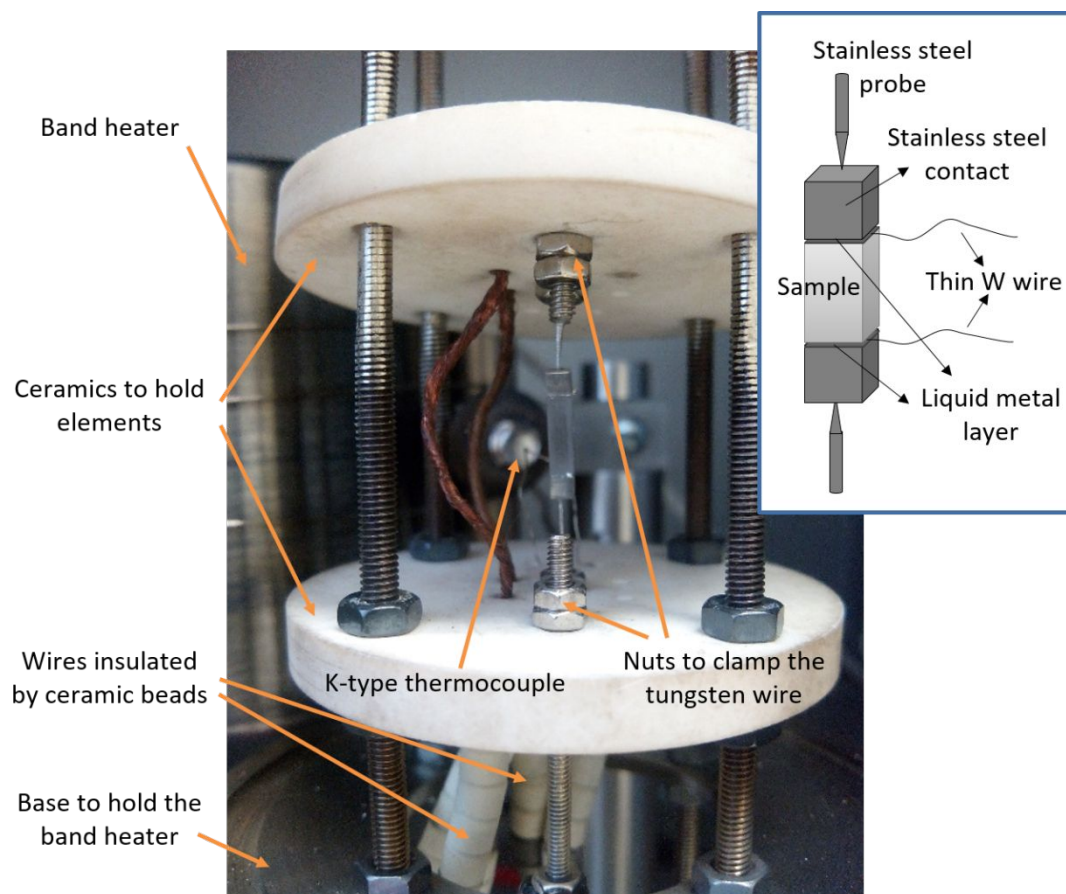
However, the efficiency of current TE materials is still limited. The search for more efficient materials is guided by the optimization of three properties, the electrical conductivity σ , the Seebeck coefficient S , and the thermal conductivity λ (which is the addition of the lattice thermal conductivity and the electronic thermal conductivity). These define a dimensionless figure of merit $zT = \sigma S^2 T / \lambda$, T is the absolute temperature, which is related to the materials efficiency.⁵ zT is typically obtained by the independent measurement as a function of temperature of σ , S , and λ . This usually requires at least two different instruments. S and σ can be measured using a single apparatus, while the most frequently used method to determine λ is the laser flash technique, which provides the thermal diffusivity α . Thus, knowledge of the specific heat C_p and the mass density d is required, since $\lambda = \alpha d C_p$.⁶ The measurement of the specific heat requires an additional measurement, performed either by using the same laser flash equipment or another instrument. For the mass density, an Archimedes balance is frequently employed. The significant number of instruments required, each with their own sources of error, and the large number of measurements to be performed, makes the task of completely characterization of TE materials quite tedious and time consuming. In addition, much of the required equipment is quite expensive, and hence not always readily accessible to all researchers. All these disadvantages entail significant obstacles in the search for better TE materials, eventually affecting the attainment of a more efficient heat to electricity energy conversion.

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2 Several techniques have been developed which allow the complete characterization of TE materials.⁷⁻¹¹ Here a
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4 new method is proposed, which unlike the previously reported techniques does not require two identical samples
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6 nor the creation of a temperature gradient across the sample, the measurement of which can introduce significant
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8 uncertainty.¹² In addition, the determination of the TE properties does not require a series of measurements, as all
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10 properties can be extracted from a single measurement. To our knowledge, this is the first time that all these
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12 advantages are offered by a measurement technique. The method is based on the measurement of the impedance
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14 signal of a TE sample that is sandwiched by a material of known thermal conductivity. Although the application of
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16 impedance spectroscopy to thermoelectricity dates back from the 2000s,^{13,14} this approach was proposed by us in
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18 2014,¹⁵ and it has only been demonstrated to date in TE modules.¹⁶⁻¹⁸ Here the approach is applied to a skutterudite
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20 material, for which complete TE characterization is achieved up to 250 °C (although with some deviations in σ at
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22 the higher temperatures). The results are compared with the values of the TE properties determined using
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24 commercially available equipment, and the random and systematic errors are calculated. The fact that the new
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26 method is based on impedance spectroscopy introduces additional advantages, since it is a widely used technique
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28 in many fields of research (solar cells,^{19,20} fuel cells,²¹ supercapacitors,²² corrosion,²³ electroceramics,²⁴ etc.). For
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30 this reason, highly reliable impedance equipment exists in the market and can be found in many research institutions,
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32 which makes the method more accessible.
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36 2. EXPERIMENTAL SETUP

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39 The setup employed for the complete characterization of the skutterudite sample is shown in Fig. 1. It is similar
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41 to the setup employed in our previous work to characterize TE materials of known Seebeck coefficient.²⁵ Unlike
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43 the previously reported setup, the TE sample ($\text{CoSb}_{2.75}\text{Sn}_{0.05}\text{Te}_{0.20}$ skutterudite²⁶ of 1.85 mm x 2.13 mm x 6.95 mm)
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45 is here sandwiched by two stainless steel (AISI 304) contacts of the same cross-sectional area and 2 mm thickness.
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47 A four-probe arrangement is employed (see inset of Fig. 1), where the current is injected and extracted by two
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49 sharpened stainless steel screws and the voltage is measured across the sample by inserting very thin (15 μm
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51 diameter, Alfa Aesar) tungsten wires at the junctions. These wires are used instead of the Cu wires employed in our
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53 previous study since reactions with the stainless steel were observed for copper at higher temperatures. In order to
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1
2 minimize the electrical and thermal contact resistances, a layer of a liquid metal (GaInSn, Alfa Aesar) is
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4 homogeneously spread at the junctions (see inset of Fig. 1).
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37 Fig. 1. Photograph of the sample holder employed. A schematic description of how the sample is contacted is provided in the inset.

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39 The two stainless steel screws which drive the current are held by nuts at holed ceramics (Macor, Corning)
40 which provide electrical insulation, as shown in Fig. 1. These screws are connected to thick copper wires insulated
41 by ceramic beads. Stainless steel screws were chosen due to their low thermal conductivity ($\approx 14 \text{ W/K} \cdot \text{m}^{-1}$), which
42 reduces heat losses by conduction. They were also sharpened for the same purpose. The very thin tungsten wires
43 that measure the potential difference are clamped at the sample holder by two nuts screwed with stainless steel
44 screws, which are held by the ceramic plates (see Fig. 1). These screws are also connected to thick copper wires
45 insulated by ceramic beads. The bottom holed ceramic disc is fixed at four threaded studs by nuts, while the top
46 ceramic is free to move to be able to allocate samples of different lengths, and additionally provide pressure to the
47 contacts. A stainless steel base is also held by nuts at the studs. This base is used to hold a band heater (Ref.
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1 MB2E2JN1-B12, Watlow) which surrounds the sample holder and is used to provide different ambient
2 temperatures. The ambient temperature is measured by a K-type thermocouple (RS) placed close to the TE sample
3 (see Fig. 1), whose temperature is controlled by a temperature controller (Watlow EZ Zone PM) which powers the
4 heater.
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11 All the impedance measurements were performed inside a stainless-steel vacuum chamber at pressure values
12 $<10^{-4}$ mbar in order to eliminate convection heat losses. In addition, the metallic vacuum chamber also serves as a
13 Faraday cage, which reduces electromagnetic noise during the measurements. The TE sample used in this study
14 was an isotropic n-type skutterudite ($\text{CoSb}_{2.75}\text{Sn}_{0.05}\text{Te}_{0.20}$), which was cut with a diamond saw of 0.3 mm diameter
15 from an original disc shape. A careful and suitable cutting is important to obtain a crack free sample of highly
16 uniform cross-sectional area. The skutterudite sample was characterized using commercial equipment in its disc
17 shape before performing the impedance measurements. A Linseis LSR-3 equipment was used to determine the
18 electrical resistivity and the Seebeck coefficient. For the thermal conductivity a Netzsch LFA 447 laser flash
19 apparatus was employed. The specific heat of the sample was determined using the same equipment via a
20 comparative method using a Pyroceram reference sample. The density of the sample, which is also required for the
21 determination of the thermal conductivity by the laser flash method, was measured using an Archimedes balance.
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35 A PGSTAT30 potentiostat (Metrohm Autolab B.V.) equipped with a FRA2 impedance module and a
36 BOOSTER10A, was used to perform the impedance spectroscopy measurements. The potentiostat was controlled
37 by the Nova 1.11 software. At each temperature the impedance measurement was conducted in 40 logarithmically
38 distributed frequency steps between 5 mHz and 500 Hz. The measurements were performed using a maximum
39 integration time of 10 s and 2 minimum integration cycles. The fitting to the impedance spectra were performed
40 using Zview software. In our previous paper it was discussed the use of the current booster to reduce a systematic
41 jump in the real impedance produced due to a change in the gain of the equipment, which occurs at frequencies
42 around 25 Hz. Although this jump can be significantly reduced if measurements are performed in the largest possible
43 current range, it distorts the spectra and due to this the fittings are performed discarding the points of frequencies
44 higher than that of the discontinuity.
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3. RESULTS AND DISCUSSION

3.1. The equivalent circuit

To obtain all the TE properties from the impedance data obtained in this work, the experimental spectra were fitted using the equivalent circuit corresponding to a TE material sandwiched between two metallic contacts.¹⁵ This equivalent circuit consists of an ohmic resistance R_{Ω} connected in series with the parallel combination of a constant temperature Warburg Z_{WCT} and an adiabatic Warburg Z_{Wa} . Each of these elements are given by,

$$R_{\Omega} = \frac{\rho_{TE} L_{TE}}{A}, \quad (1)$$

$$Z_{WCT} = R_{TE} \left(\frac{j\omega}{\omega_{TE}} \right)^{-0.5} \tanh \left[\left(\frac{j\omega}{\omega_{TE}} \right)^{0.5} \right], \quad (2)$$

$$Z_{Wa} = R_C \left(\frac{j\omega}{\omega_C} \right)^{-0.5} \coth \left[\left(\frac{j\omega}{\omega_C} \right)^{0.5} \right], \quad (3)$$

where ρ_{TE} , L_{TE} , and A are the electrical resistivity, length, and cross-sectional area of the TE material, respectively, $j^2 = -1$, ω is the angular frequency, and ω_{TE} and ω_C are the characteristic angular frequencies of thermal diffusion in the TE sample ($\omega_{TE} = \alpha_{TE} / (L_{TE}/2)^2$; α_{TE} denoting the thermal diffusivity of the TE material) and in the contact ($\omega_C = \alpha_C / (L_C)^2$; α_C denoting the thermal diffusivity of the contact). R_{TE} is the TE resistance,²⁷ and R_C is a TE resistance induced by the contact. They are given by,

$$R_{TE} = \frac{S^2 T L_{TE}}{\lambda_{TE} A}, \quad (4)$$

$$R_C = 2 \frac{S^2 T L_C}{\lambda_C A}, \quad (5)$$

where λ_{TE} and λ_C are the thermal conductivity of the TE material and the stainless steel contact, respectively, and L_C the length of the latter.

From the curve fits, R_{Ω} , R_{TE} , R_C , ω_{TE} , and ω_C can be obtained. Hence, using Eq. (1), the electrical resistivity can be determined as,

$$\rho_{TE} = \frac{R_{\Omega}A}{L_{TE}}. \quad (6)$$

From Eq. (5), the Seebeck coefficient can be obtained as,

$$S = \sqrt{\frac{R_C \lambda_C A}{2TL_C}}. \quad (7)$$

It should be noted that in order to determine S in Eq. (7), the thermal conductivity of the stainless steel contact is required, for which literature values may be used.²⁸ Combining Eq. (1) and Eq. (4),

$$zT = \frac{R_{TE}}{R_{\Omega}}. \quad (8)$$

3.2. Characterization by the impedance method

Five cycles were performed on the skutterudite sample, each cycle comprising a set of five impedance measurements at different temperatures (50, 100, 150, 200 and 250 °C). Before the beginning of each cycle, the sample was newly assembled with fresh contacts. In order to obtain accurate impedance results, it is important to establish a suitable current amplitude for the measurements (the lowest amplitude possible with non-noisy measurements). This is to minimize the influence of non-linear effects such as the Joule heating and the variation of the TE properties with temperature, as discussed in our previous papers.^{25,29} Hence, before performing the cycles, impedance spectroscopy measurements at different current amplitudes (40, 60, 80, 100 and 120 mA) were performed at each temperature in order to identify their optimal values. Fig. 2 shows the experimental impedance spectra and the corresponding fits for one of the five cycles measured. Fitting errors below 0.1, 0.5, and 12% were obtained for R_{Ω} , R_{TE} , and R_C , respectively. It can be observed that even for the spectrum at 50 °C, where the skutterudite shows lower performance and the impedance values are very small, the impedance response is clearly observed. In any case, the measured points differ in just few tenths of $\mu\Omega$ at high frequencies (bottom left part), and

a powerful impedance analyzer (with high resolution and accuracy) is required in order to obtain sensitive measurements.

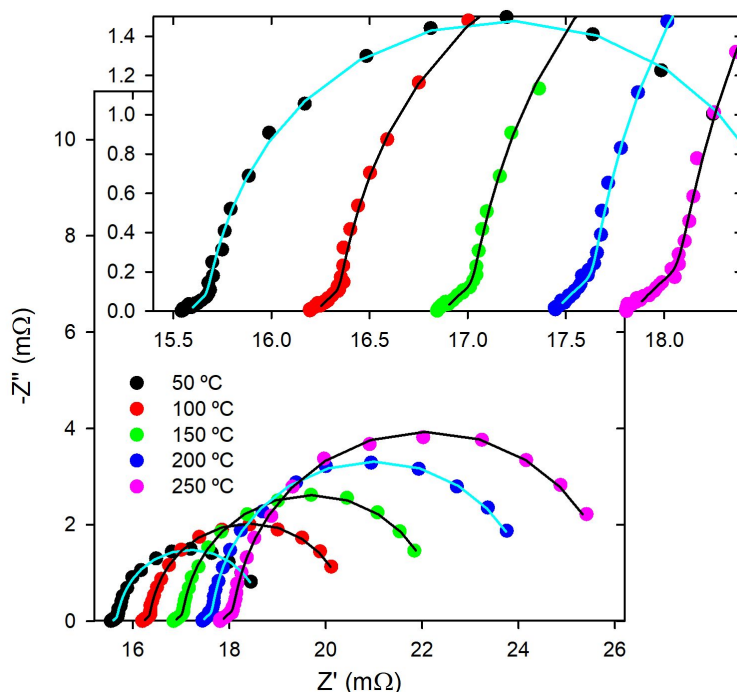


Fig. 2. Impedance spectroscopy measurements at different temperatures from one of the five measurement cycles performed. The dots represent the experimental values and the lines represent the fit to these data. The inset shows the magnification of the high frequency part.

The TE properties were obtained from Eq. (6) to Eq. (9) using the average values of R_{Ω} , R_{TE} , and R_C from the five measurements at each temperature. The thermal conductivity of the contact λ_C (stainless steel AISI 304), which was needed for the Seebeck coefficient determination, was obtained from,²⁸

$$\lambda_C = 10.33 + 15.4 \times 10^{-3}T - 7.0 \times 10^{-7}T^2. \quad (9)$$

The validity of Eq. (9) was verified by performing measurements of the stainless steel AISI 304 thermal conductivity by a laser flash apparatus (LFA 467 HT from Netzsch) up to 150 °C. The deviations found with respect to the equation were lower than 2.7%. It is important in order to clearly discern the 45° straight line feature at high frequencies (see bottom part of the inset of Fig. 2) that λ_C is around an order of magnitude higher than λ_{TE} , otherwise this feature will overlap with the semicircle part.

1 Fig. 3 shows the TE properties obtained by the impedance spectroscopy method compared with results from
2 commercial equipment. All the properties show a good agreement with the commercial equipment measurements,
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4 except the electrical resistivity [Fig. 3(a)], which shows slightly higher values (around 6%), which is due to the
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6 contribution from the contact resistance, which is not completely suppressed since the W wires are inserted at the
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8 junctions. It is known that GaInSn liquid metal can have nm-length native oxide layers at its surface, which can
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10 impact its wetting behavior and electrical resistivity. This might contribute to the higher electrical resistivity values
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12 found.³⁰ It can be also observed for this property that as the temperature increases the error bars become larger and
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14 the trend slightly deviates from the behavior found with the commercial equipment. This is related to the fact that
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16 the GaInSn liquid metal tends to react with the TE sample at around 250 °C. This is the limiting constraint on the
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18 maximum temperature of operation, since the rest of the elements of the setup can stand far higher temperature
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20 values. Hence, if a suitable solder or liquid metal for the sample to be measured were found, this method could
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22 increase its capability at higher temperatures. These aspects mentioned for the electrical resistivity also influence
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24 the zT [Fig. 3(d)] due to Eq. (8), which exhibits larger errors and deviations at the highest temperatures.
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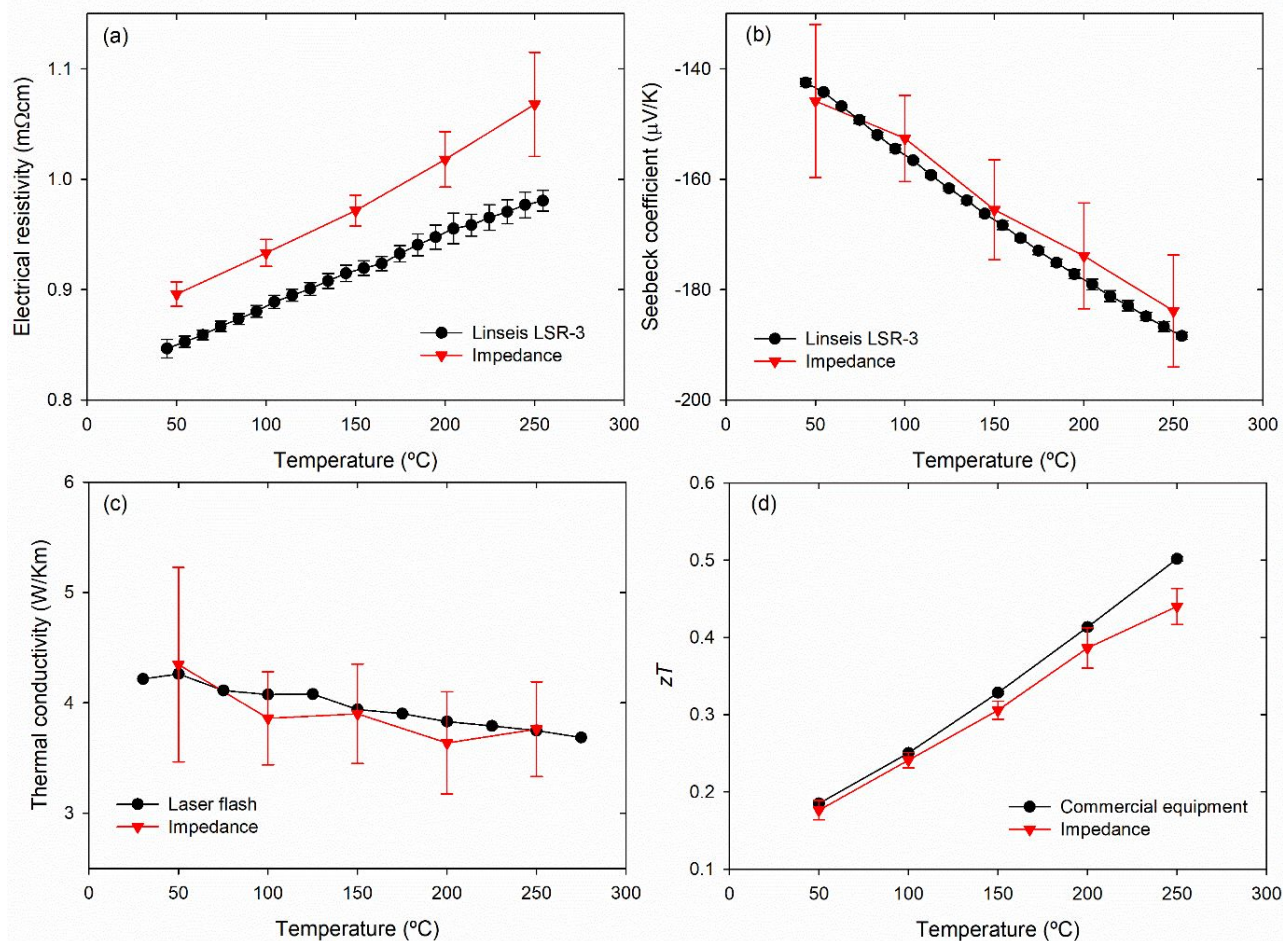


Fig. 3. (a) Electrical resistivity, (b) Seebeck coefficient, (c) thermal conductivity, and (d) zT values extracted from the impedance method and compared with results from different commercial equipment. The error bars account for the total combined random errors (u_c), excluding the contribution from the specific heat for the laser flash case. The confidence interval is 1σ .

3.3. Precision and accuracy evaluation

In order to quantify the precision and accuracy of the impedance method, random and systematic errors, respectively, were calculated for all the determined TE properties. The total combined random errors u_c of each property were obtained using,³¹

$$u_c^2 = \sum_{i=1}^N \left(\frac{\partial f}{\partial x_i} \right)^2 u^2(x_i), \quad (10)$$

being f each of the TE properties (S , λ_{TE} , ρ_{TE} , or zT), and x_i each of the parameters with an associated error u .

The random errors for the Seebeck coefficient were calculated taking into account (i) the standard deviation of the

1 five measurements at each temperature to obtain the average value of R_C , (ii) the mean fitting error of the five R_C
2 measurements at each temperature, (iii) the uncertainty in the area of the sample, (iv) the uncertainty of the
3 thermocouple ($u(T)=1$ °C), and (v) the uncertainty in the length of the contacts, which was measured using a caliper
4 ($u(L_C)=0.005$ mm). The contribution of the thermal conductivity of the contact was neglected. From all the above
5 contributions, (i) and (ii) were the most significant compared to the others, which can be considered negligible.
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13 The random errors for the thermal conductivity were calculated taking into account (i) the uncertainty of the
14 Seebeck coefficient ($u_c(S)$), (ii) the uncertainty of the thermocouple ($u(T)=1$ °C), (iii) the uncertainty in the length
15 of the sample ($u(L_{TE})=0.005$ mm), (iv) the uncertainty in the area of the sample, and (v) the standard deviation of
16 the five measurements at each temperature to obtain the average value of R_{TE} . The contribution from the fitting
17 errors in R_{TE} (which were $<0.5\%$) was discarded since it was negligible in comparison with the standard deviation.
18 From all the contributions considered, the uncertainty in the Seebeck coefficient and the standard deviation of R_{TE}
19 are the most significant, the Seebeck contribution being an order of magnitude higher. Hence, the precision in the
20 thermal conductivity determination is strongly influenced by the precision in the Seebeck coefficient measurement.
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31 The random errors for the electrical resistivity were calculated taking into account (i) the uncertainty in the
32 length of the sample, (ii) the uncertainty in the cross-sectional area, and (iii) the standard deviation from the five
33 measurements at each temperature to obtain the average R_Ω . It should be noticed that the latter contribution is the
34 most significant, since it is around two orders of magnitude larger than the others. As occurred for R_{TE} , the
35 contribution of the fitting errors for R_Ω ($<0.1\%$) was neglected. Finally, the random errors for zT were calculated
36 from the contributions of the standard deviations of both R_Ω and R_{TE} . The error bars shown in Fig. 3 correspond to
37 the calculated random errors for each property, which are also shown in Table 1. Most of the random errors are
38 $\approx 5.5\%$, $<13\%$, $<2.5\%$, and between 4 and 7% for S , λ_{TE} , ρ_{TE} , and zT , respectively, which demonstrates the good
39 precision of the method, although the thermal conductivity is less precise due to the quadratic dependence on the
40 Seebeck coefficient [see Eq. (7)]. At 50 °C higher values are found for S and λ_{TE} due to a lower degree of repeatability
41 at this temperature in one of the 5 cycles performed.
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Table 1. Average values with their associated random, systematic and total errors of the thermoelectric properties of the skutterudite sample obtained by the impedance spectroscopy method.

| | Temperature (°C) | Mean value | Systematic error (%) | Random error (%) | Total error (%) |
|---|------------------|------------------------------------|----------------------|------------------|-----------------|
| Seebeck coefficient (S) | 50 | -145.8 μVK^{-1} | 1.08 | 9.51 | 9.57 |
| | 100 | -152.6 μVK^{-1} | 1.89 | 5.11 | 5.45 |
| | 150 | -165.5 μVK^{-1} | 1.13 | 5.48 | 5.60 |
| | 200 | -173.9 μVK^{-1} | 2.34 | 5.51 | 5.99 |
| | 250 | -183.9 μVK^{-1} | 2.03 | 5.51 | 5.87 |
| Thermal conductivity (λ_{TE}) | 50 | 4.35 $\text{WK}^{-1}\text{m}^{-1}$ | 2.00 | 20.26 | 20.36 |
| | 100 | 3.86 $\text{WK}^{-1}\text{m}^{-1}$ | 5.28 | 10.93 | 12.14 |
| | 150 | 3.90 $\text{WK}^{-1}\text{m}^{-1}$ | 0.99 | 11.54 | 11.58 |
| | 200 | 3.64 $\text{WK}^{-1}\text{m}^{-1}$ | 5.05 | 12.71 | 13.68 |
| | 250 | 3.76 $\text{WK}^{-1}\text{m}^{-1}$ | 0.34 | 11.40 | 11.41 |
| Electrical resistivity (ρ_{TE}) | 50 | 0.896 $\text{m}\Omega\text{cm}$ | 5.06 | 1.24 | 5.21 |
| | 100 | 0.933 $\text{m}\Omega\text{cm}$ | 5.53 | 1.31 | 5.68 |
| | 150 | 0.972 $\text{m}\Omega\text{cm}$ | 6.07 | 1.43 | 6.24 |
| | 200 | 1.018 $\text{m}\Omega\text{cm}$ | 7.43 | 2.47 | 7.83 |
| | 250 | 1.068 $\text{m}\Omega\text{cm}$ | 9.06 | 4.41 | 10.08 |
| Dimensionless figure of merit (zT) | 50 | 0.176 | 6.69 | 7.07 | 9.73 |
| | 100 | 0.241 | 0.04 | 4.07 | 4.07 |
| | 150 | 0.306 | 4.78 | 3.85 | 6.13 |
| | 200 | 0.386 | 1.97 | 6.77 | 7.05 |
| | 250 | 0.440 | 8.62 | 5.29 | 10.12 |

Systematic errors u_s were calculated for the TE properties considering as true values the results obtained from the commercial equipment. They are also included in Table 1. Systematic errors are <2.5%, <5.5%, between 5 and 9%, and <9% for the S , λ_{TE} , ρ_{TE} , and zT , respectively, demonstrating a good agreement with the characterization performed with commercial equipment.

Finally, the total uncertainty of the method u_T is obtained for each property as $u_T=(u_c^2+u_s^2)^{0.5}$ and also shown in Table 1. For S and λ_{TE} the total errors are predominantly <6% and <14%, respectively. For these two parameters, it is evident that the principal contribution to the total error comes from the random error, which is higher than the systematic contribution. For ρ_{TE} and zT , total errors are approximately from 5 to 10%, and from 4 to 10%, respectively. In this case the random and systematic contributions do not show the large differences as in the case of S and λ_{TE} and more equally contribute to the total error.

4. CONCLUSIONS

In summary, the ability to perform a complete characterization of all TE properties of a bulk material as a function of temperature, from a single electrical impedance spectroscopy measurement, using one apparatus is demonstrated for a low-performance TE material (skutterudite sample) of modest properties up to 250 °C. The TE properties were determined from fittings performed to the experimental impedance spectra employing a suitable equivalent circuit. Random errors were calculated by performing five measurements at each temperature remaking contacts, showing a good precision of the method ($\approx 5.5\%$, $<13\%$, $<2.5\%$, and between 4 and 7% for the S , λ_{TE} , ρ_{TE} , and zT , respectively). The random errors in the determination of thermal conductivity are higher due to the quadratic dependence of this property with the Seebeck coefficient. Systematic errors were also calculated by comparison with characterization results from commercial equipment obtained from the same sample, resulting in errors $<2.5\%$, $<5.5\%$, between 5 and 9%, and $<9\%$ for the S , λ_{TE} , ρ_{TE} , and zT , respectively, which illustrates the accuracy of the method. These results demonstrate the potential of the method as a powerful tool to significantly facilitate the task of characterization of bulk TE materials and thus the search for a more efficient heat to electricity energy conversion.

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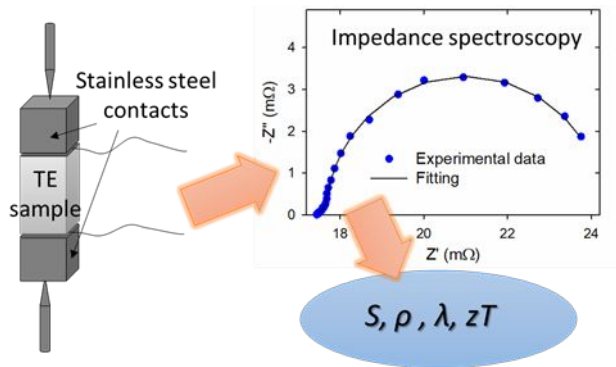
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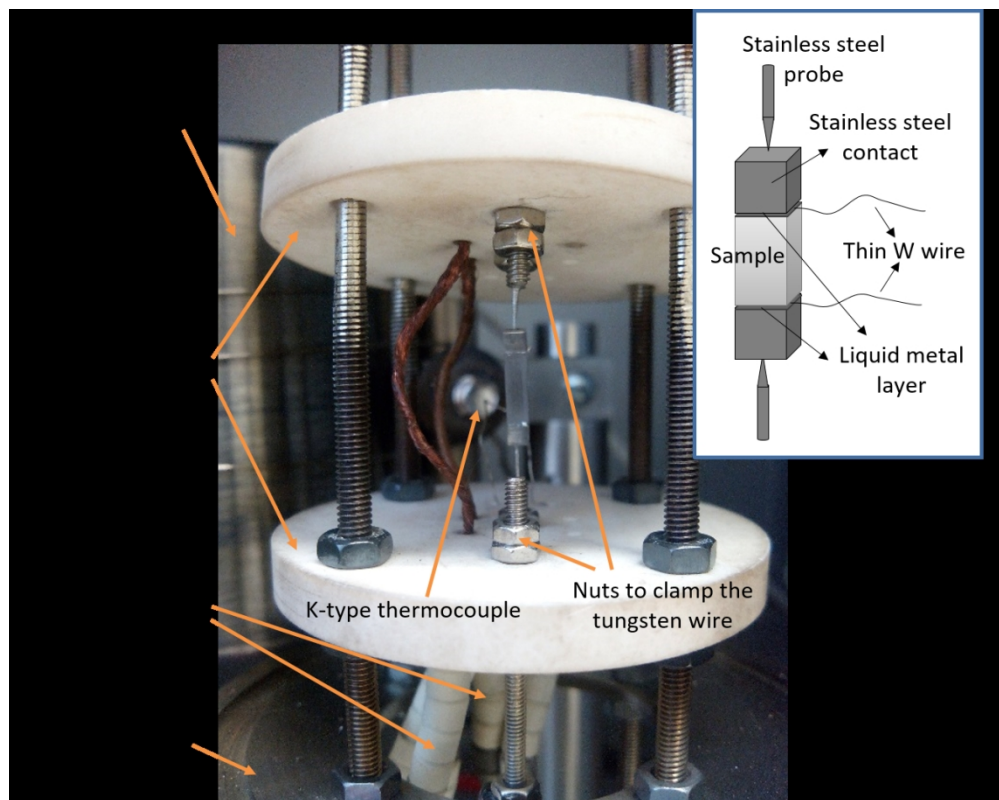
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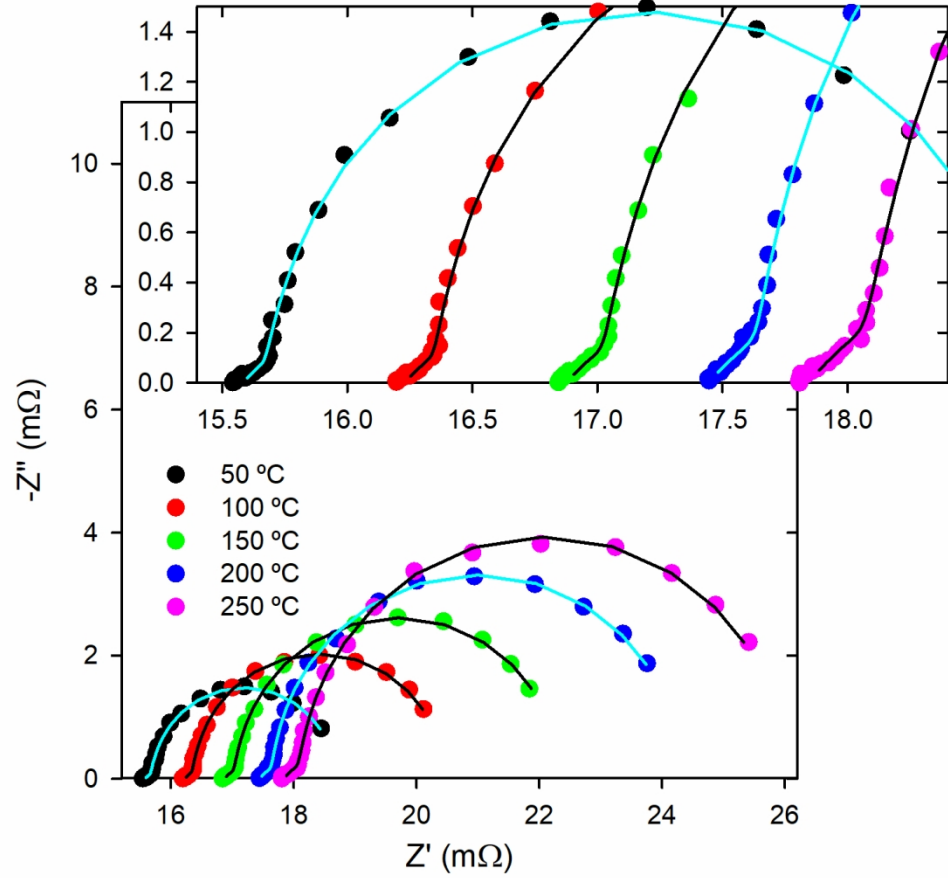
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TOC Graphic

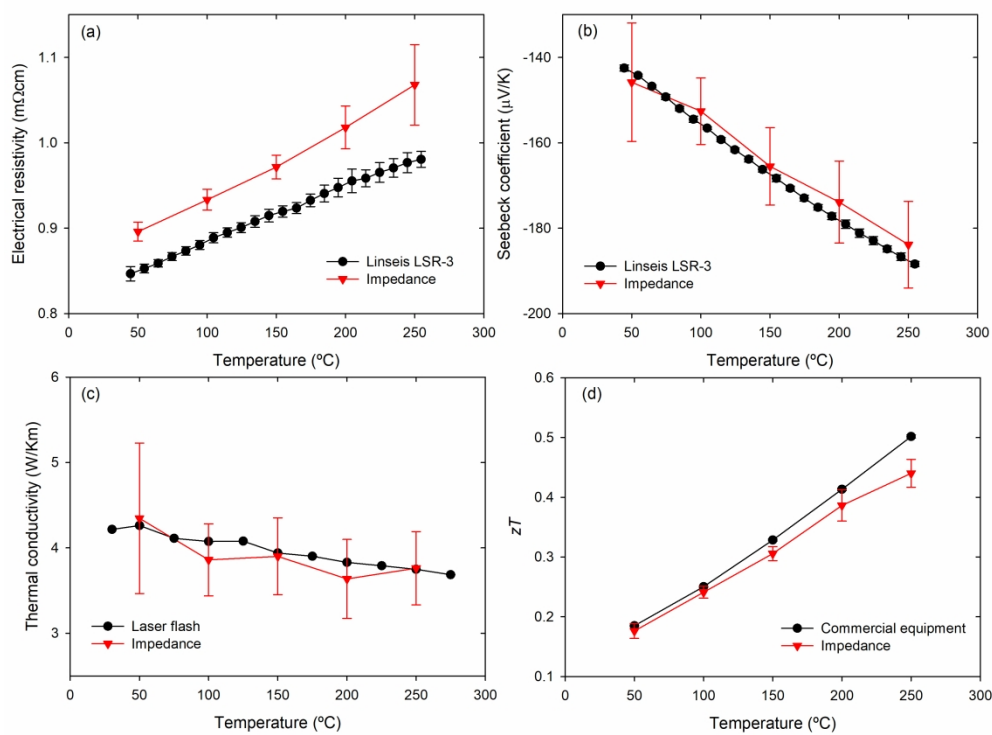




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